

Bilayer thin films based on $\text{K}_3\text{Bi}_2\text{I}_9/\text{MO}$ ($\text{M}=\text{Ti}$, Zn , and Cu) for CO_2 photoreduction

E. Luévano-Hipólito, O.L. Quintero-Lizárraga, M.R. Alfaro-Cruz, L.I. Ibarra-Rodríguez, L.M. Torres-Martínez

This work reports the fabrication of bilayer thin films of $\text{K}_3\text{Bi}_2\text{I}_9/\text{MO}$ ($\text{M}=\text{Ti}$, Cu , Zn) as an extrinsic strategy to enhance the stability of lead-free metal halide perovskite (LFMHPs) for CO_2 photoreduction. The perovskites were grown on mica substrates by the spin-coating technique, which serves as a seed layer for the deposition of TiO_2 , ZnO , and CuO oxides by physical (sputtering) and chemical (atomic layer deposition) techniques. The $\text{K}_3\text{Bi}_2\text{I}_9/\text{MO}$ samples exhibited enhanced light absorption and better stability than the reference $\text{K}_3\text{Bi}_2\text{I}_9$ perovskite. Despite the efficiency of the bilayer films was lower than the reference ($\text{K}_3\text{Bi}_2\text{I}_9$) for CO_2 photoreduction, the $\text{K}_3\text{Bi}_2\text{I}_9/\text{CuO}$ arrangement favored better stability and efficiency. These results provide an alternative to design new and more stable optical devices for photocatalytic applications.

Introduction

CO_2 photoreduction

The CO_2 reduction by heterogeneous photocatalysis is based on its conversion towards value-added products or renewable solar fuels, commonly known as “artificial photosynthesis” due to its similarity with natural photosynthesis present in the biological fixation of CO_2 [1]. This process requires CO_2 , H_2O , sunlight, and a photocatalyst (a semiconductor material) to generate solar (and renewable) fuels. The reaction mechanism of the CO_2 photoreduction starts when the photocatalyst is excited by an external energy (ideally, sunlight) of equal or higher value than its band gap energy. Then, this excitation promotes the transfer of electrons from the valence band (VB) to its conduction band (CB), simultaneously generating electrons (e^-) and holes (h^+) capable of oxidizing the adsorbed species, e.g., water to produce reducing species (H^+) that react with the CO_2 molecule to generate different oxygenated products (CO , HCOOH , HCHO , CH_3OH) and short-chain olefins (C_2H_4 , C_3H_6) [2]. The efficiency and the selectivity of this reaction will depend on several factors, among which the type of photocatalyst stands out [3], which includes oxide and non-oxide, sulfides, molecular catalysts, and other/diverse catalysts [4]. Among them, the materials with perovskite structures, mainly metal halide perovskites, had promoted high efficiencies for CO_2 reduction, mainly to produce CO and CH_4 [5, 6].

Lead-free metal halide perovskites

Metal halide perovskites (MHP) are recognized as an ideal choice for CO_2 photoreduction owing to their outstanding optoelectronic properties such as light absorption coefficient, tunable bandgap, superior defect tolerance, long carrier diffusion length, and remarkable quantum efficiency. These favorable properties have been used to design different photosystems, e.g., photovoltaic, photodetectors, light-emitting diodes, and, recently, as photocatalysts. Remarkably, the application of MHP in solar cells has shown the rapid improvement of the power conversion efficiency from 3.8 to 25.2% in the last years [6]. Inspired by the excellent properties of MHPs, the study of their photocatalytic activity for CO_2 reduction to generate renewable solar fuels has recently attracted much attention.

So far, cesium lead bromide (CsPbBr_3) is the MHP with the highest yield reported for CO and CH_4 production ($431 \mu\text{mol g}^{-1} \text{h}^{-1}$) of this group of materials [7]. Despite the

promising activity of MHPs for CO_2 photoreduction, lead in its structure entails environmental toxicity problems. Thus, numerous efforts have been made to replace this metal without sacrificing its good performance by developing lead-free metal halide perovskites (LFMHPs). One example is the replacement of Pb by trivalent cations that allow the formation of new stoichiometries, such as $\text{A}_3\text{B}_2\text{X}_9$ ($\text{A}=\text{Cs}$, K , $\text{B}=\text{Bi}$, Sn , $\text{X}=\text{I}$, Br , Cl) stoichiometry. Among these materials, the use of the Cs^+ and Bi^{3+} cations in the A and B sites, respectively has resulted in outstanding efficiencies for CO and CH_4 production from CO_2 photoreduction, with the $\text{Cs}_3\text{Bi}_2\text{I}_9$ perovskite [8]. Recently, our research group demonstrated that alternative formulations are possible by the replacement of Cs^+ with K^+ to give place to the formation of $\text{K}_3\text{Bi}_2\text{I}_9$ perovskites with promising activity to produce liquid solar fuels, e.g., HCOOH from CO_2 photoreduction [9]. Although significant efforts have been devoted to designing more effective photocatalysts for CO_2 photoreduction, some issues must be solved before the application of LFMHPs, e.g., charge recombination, instability, and band engineering, to control the selectivity of the reaction.

The band structure of a semiconductor photocatalyst intrinsically controls its CB and VB potential and, thus, influences its activity for different photocatalytic reactions [10]. One method to adapt the band structure is the heterostructure engineering, incorporating additional semiconductors to the photocatalyst. The junction between two or three semiconductors significantly impacts light

E. Luévano-Hipólito , M.R. Alfaro-Cruz 

SECIHTI-Universidad Autónoma de Nuevo León, Facultad de Ingeniería Civil- Laboratorio de Ecomateriales y Energía
San Nicolás de los Garza, N.L., 66455 México.

O.L. Quintero-Lizárraga , L.M. Torres-Martínez

Universidad Autónoma de Nuevo León, Facultad de Ingeniería Civil-Laboratorio de Ecomateriales y Energía
San Nicolás de los Garza, N.L., 66455 México.

L.I. Ibarra-Rodríguez, L.M. Torres-Martínez 

Centro de Investigación en Materiales Avanzados, S. C. (CIMAV)
Chihuahua, Chih., 31136, México.

Received: October 29th, 2024.

Accepted: December 23rd, 2025.

Published: January 12th, 2026.

© 2026 by the authors. Creative Commons Attribution

https://doi.org/10.47566/2026_syv39_1-260101

absorption, charge transfer efficiency, active sites, the affinity for CO₂ adsorption, and the available redox potentials to control the selectivity of the reaction. Different heterostructures with higher efficiencies than bare materials have been reported with LFMHPs, e.g., K₃Bi₂I₉/BiOI [9], Cs₃Bi₂I₉/Bi₂WO₆ [11], Cs₃Bi₂I₉/CeO₂ [12], Cs₃Bi₂Br₉/In₄SnS₈ [13], and other double LFMHPs as Cs₂AgBiBr₆/TiO₂ [14]. Also, heterostructure engineering is an efficient strategy for encapsulating the perovskite and improving its efficiency [15]. In this context, the use of simple oxides could represent a good option for designing heterostructures with higher stability in aqueous systems; for example, ZnO is considered a layered window in solar cells to protect the internal semiconductors and, in some cases, could increase the light absorption and improve the transport of the photogenerated charges [16], while CuO has high CO₂ affinity and stability for CO₂ photoreduction [17]. Considering these factors, this work proposes the design of bilayer heterostructures based on K₃Bi₂I₉ with TiO₂, ZnO, and CuO to analyze their effect on CO₂ photoreduction efficiency and stability in aqueous solution. The perovskite was deposited by spin-coating on mica substrates; meanwhile, the simple oxides (or encapsulants) by sputtering since a technique that allows good homogeneity on the films.

Experimental

Deposition of K₃Bi₂I₉ films

The deposition of the perovskite films was performed by spin-coating. In the first step, an ink of the precursors was synthesized according to a previous report [9]. The method consists of mixed stoichiometric (3:1 molar) amounts of KI (Fermont, 99%) and BiI₃ (Aldrich, 99%) in N,N-dimethylformamide. The ink was sonicated for 15 min and stirred under a magnetic stirrer at 70 °C for 2 h. Then, it was deposited over mica substrates by spin-coating at 2000 rpm for 30 s and annealed at 100 °C for 5 min (Figure 1a).

Encapsulation of K₃Bi₂I₉ with metal oxide films

TiO₂, CuO, and ZnO were deposited separately on the K₃Bi₂I₉ films by two techniques. A layer of TiO₂ or CuO was grown over the K₃Bi₂I₉ surface by DC magnetron sputtering using a metallic Cu and Ti target (99.995% purity, 2.00"×0.250", Kurt J. Lesker). The background pressure of the sputtering chamber was 8×10⁻⁶ Torr, and the working pressure during deposition was 1.5×10⁻² Torr. The chamber was filled with argon gas (99.999% purity) and the power used during deposition was 30 W for each target. Afterwards, the bilayer films were calcined at 400 °C for 2 h to promote the oxidation of the metals into CuO and TiO₂ (Figure 1b).

On the other hand, ZnO was deposited by atomic layer deposition (ALD) at 80 °C. The precursors used were DEZ (Diethyl Zinc 95%, STREM Chemicals, CAS 557-20-0) and deionized water. For ZnO deposition it was necessary 500 cycles with the following conditions: 250 ms DEZ, 2 s purge, 250 ms H₂O, and 2 s purge. The registered pressure was 0.9 mbar; these conditions favored a thickness around 100 nm. The methodology used for the deposition and encapsulation of the K₃Bi₂I₉ films is shown in Figure 1c.

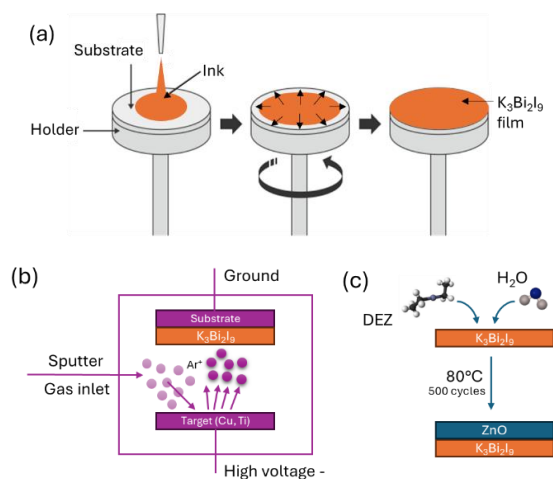


Figure 1. Methodology for the deposition of K₃Bi₂I₉ films (a) by spin-coating and their encapsulation by (b) sputtering and (c) atomic layer deposition techniques.

Characterization

The morphological characterization was achieved using Scanning Electron Microscopy (SEM) in a JEOL JSM-6490LV microscope. UV-Visible diffuse reflectance spectroscopy (DRS) was employed to calculate the bandgap energy of the films using an Agilent Technologies UV-Vis-NIR Cary 5000 spectrophotometer equipped with an integrating sphere. The crystal phase was confirmed by X-ray diffraction (XRD) using a PANalytical diffractometer with Cu-Kα radiation.

CO₂ photoreduction assays

The activity of the thin films for CO₂ photoreduction was evaluated at room temperature in a Pyrex glass reactor. For these experiments, one film of 5 cm² was used as catalyst. It is worth mentioning that the mass on each film was lower than 0.01 g. The reactor was filled with 100 mL of distilled water under continuous stirring. The reactor was saturated with continuous CO₂ flow and then, a visible lamp (> 400 nm) was turned on outside the reactor. The spectrum of the visible lamp used was previously reported by our research group [18]. A continuous flow (1 mL min⁻¹) was maintained during the reaction. The products were monitored using gas and liquid chromatography by Thermo Scientific Trace 1310 and Shimadzu Prominence-i LC-2030C model.

Results

Characterization of the films

The morphology of the films was studied by Scanning Electron Microscopy. The mica substrate showed a smooth surface with negligible roughness while the reference perovskite exhibited a compact microstructure with a certain degree of agglomeration, as seen in Figures 2a and 2b. The bilayer films exhibited different microstructures showing that TiO₂ and ZnO do not favor a complete coverage of the perovskite (Figure 2, c-d). As in the case of TiO₂, some nanosheets were observed, which could be related to the presence of a partial decomposition of the base material [9].

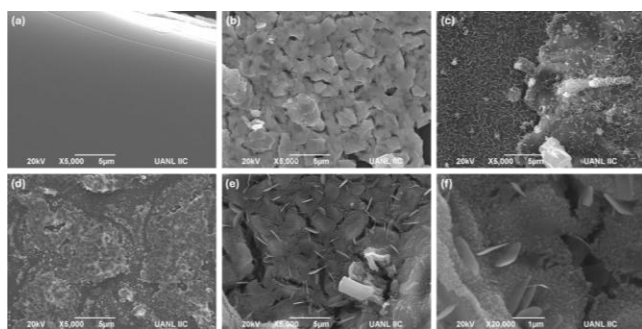


Figure 2. SEM images of (a) mica, (b) $K_3Bi_2I_9$, (c) $K_3Bi_2I_9/TiO_2$, (d) $K_3Bi_2I_9/ZnO$, and (e, f) $K_3Bi_2I_9/CuO$ films.

The CuO bilayer exhibited better coverage with small nanospheres that grew over the perovskite layers (Figure 2, e-f), which can benefit the photocatalytic reactions.

Figure 3 shows the absorption spectra of the different films, with different absorption bands observed for each sample. Notably, TiO_2 and ZnO showed absorption bands at higher wavelengths than usual, probably due to the absorption of the mica substrate. Although the simple oxides deposited on mica substrates absorb energy from 500 nm, showing their characteristic bands around 350 nm (TiO_2 and ZnO) and 700 nm (CuO). The absorption spectra of the bilayer films exhibited similar bands, as seen in Figure 3b. The band gap of the reference $K_3Bi_2I_9$ film was 1.9 eV, which agrees with previous reports (9), while the band edge of the reference film shifted to higher wavelengths, which evidenced an enhanced light absorption. This phenomenon was more evident in the $K_3Bi_2I_9/CuO$ film.

Figure 4 shows the XRD pattern of the substrate and the reference $K_3Bi_2I_9$ film. The substrate exhibited an intense and sharp reflection around $2\theta \sim 10^\circ$ associated with the muscovite phase (card 07-0042) of the mica. After the deposition of the $K_3Bi_2I_9$ film, a small reflection located $\sim 13^\circ$ in 2θ is associated with the monoclinic phase of this material. It is important to mention that the diffractogram of the bilayer films did not show additional reflections of the simple oxides, probably due to the low thickness of the film. Previous studies under similar experimental conditions have reported that metal oxide films with thicknesses below 100 nm typically yield signals too weak to be detected by XRD [19–22].

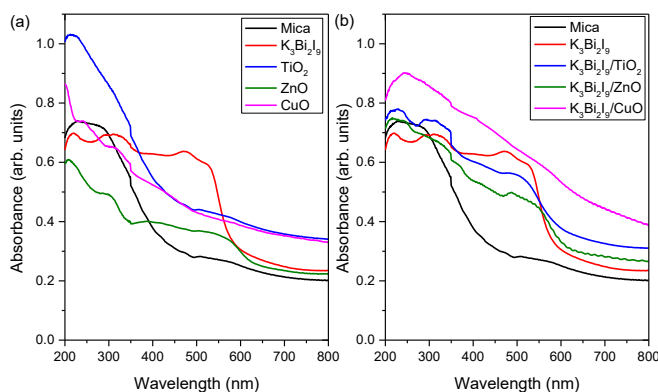


Figure 3. Absorption spectra of the (a) reference and (b) bilayer films grown on mica substrates.

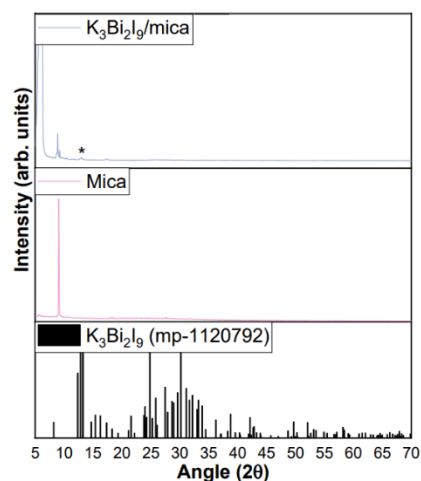


Figure 4. X-ray diffraction of the $K_3Bi_2I_9$ films on mica substrate.

Photocatalytic activity

The references and the bilayer films were evaluated as catalyst in CO_2 photoreduction at room conditions (298 K, 1 bar). Different products such as $HCOOH$, $HCHO$, CH_3OH , CH_4 and CO were monitored during this reaction, however, only formic acid was detected. Figure 5a shows the amount of formic acid generated after 1 h of reaction with the references.

To confirm the stability of each system, the evolution of $HCOOH$ was monitored by one hour of reaction under visible-light irradiation. Figure 6a shows that the reference sample exhibited rapid activation; however, its efficiency for $HCOOH$ production reached equilibrium faster than the other samples. On the contrary, the $K_3Bi_2I_9/CuO$ film exhibited a linear trend and a more constant fuel production, while the rest of the samples showed low activity. The reference and the bilayer film with the highest activity was evaluated in five consecutive cycles to observe changes and differences in the $HCOOH$ yield. As shown in Figure 6b, the activity of the $K_3Bi_2I_9$ perovskite significantly decreased after the second cycle of evaluation, which confirms its low stability in aqueous medium under irradiation. Alternatively, despite the bilayer $K_3Bi_2I_9/CuO$ film shows lower activity than the bare sample, it exhibited a more stable production of formic acid. These results confirm that the construction of bilayer films is a suitable strategy to delay the degradation of the perovskite in an aqueous medium.

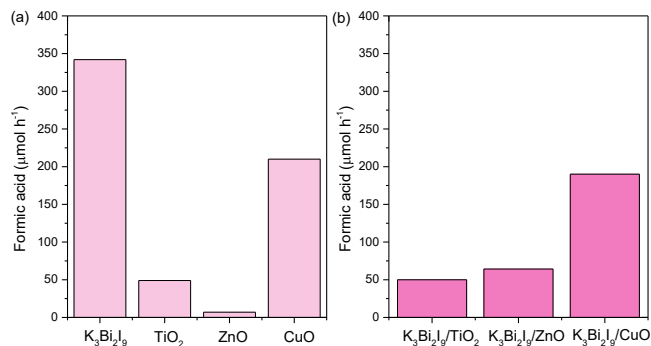


Figure 5. $HCOOH$ production from CO_2 reduction using (a) references and (b) bilayer films under visible light. Catalyst area = 5 cm^2 , mass < 0.01 g.

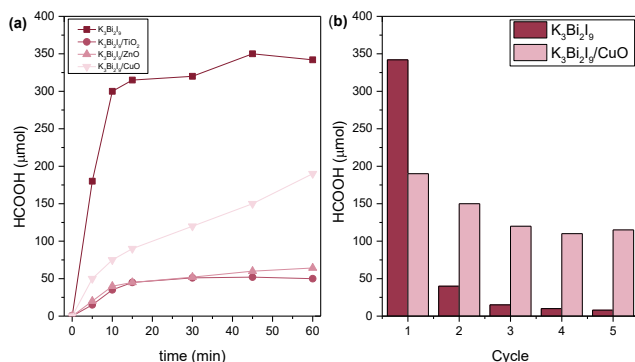


Figure 6. (a) Evolution of HCOOH production using the reference and bilayer films and (b) comparison of the HCOOH yield in 5 consecutive cycles of evaluation under visible light (area = 5 cm², mass < 0.01 g).

It is well recognized that rigorous control experiments are essential to rule out possible pseudo-catalytic contributions for the K₃Bi₂I₉/CuO film. In this study, dark, and Ar-purged control experiments were performed to confirm that the observed activity arises from the films under illumination to validate the origin of the detected products. A clear signal was obtained only under CO₂ under irradiation, while negligible amounts were observed in the dark or under Ar atmosphere (< 20 μmol). These findings confirm that the detected formic acid arises from the photocatalytic CO₂ reduction process rather than from side reactions or impurities.

Discussion of the mechanism

The changes on the photocatalytic activity of the reference and bilayer films on the CO₂ reduction would be based on the band engineering of each oxide/halide. To support this hypothesis, the redox potentials of the conduction (CB) and valence bands (VB) were calculated theoretically and a charge transfer mechanism on each system are shown in Figure 8. When the oxides with wider band gaps, e.g., TiO₂ and ZnO, are used as window layers, the transfer of the photogenerated charges to the perovskite could be hindered due to the mismatch of the band structure between the semiconductors (Figure 8a). On the other hand, the K₃Bi₂I₉/CuO film presented a better match between the

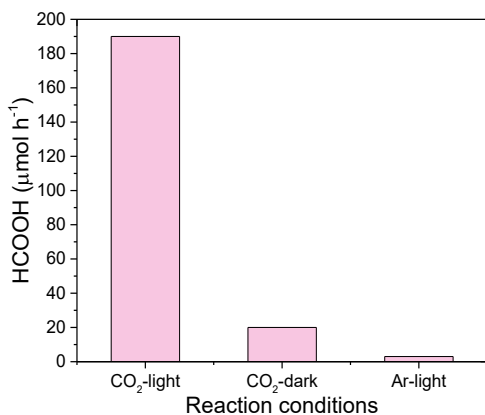


Figure 7. Control experiments (dark and Ar-purged) of the system with the best stability K₃Bi₂I₉/CuO film.

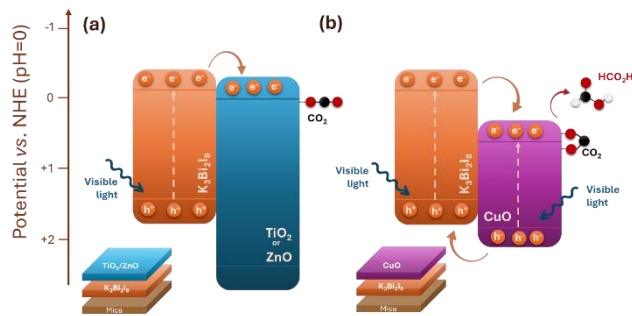


Figure 8. Proposed mechanism for CO₂ reduction using K₃Bi₂I₉ perovskites encapsulated with (a) ZnO or TiO₂ and (b) CuO oxides.

semiconductors, favoring the electron transfer from the perovskite CB to the oxide (CuO) CB; meanwhile, the holes can be transferred from the VB of CuO to the K₃Bi₂I₉ VB (Figure 8b). In addition, the surface characteristics of the CuO film may enhance CO₂ adsorption through uni- and bidentate bonds [23, 24]. This property could promote stronger interactions with CO₂ molecules, thereby facilitating higher conversions into value-added products, such as HCOOH.

The higher selectivity for formic acid generation from the photocatalytic CO₂ reduction using K₃Bi₂I₉ perovskite materials, particularly with the addition of CuO, can be attributed to several factors related to the properties of these materials:

- Perovskite materials, such as K₃Bi₂I₉, have unique structural characteristics that allow for the efficient adsorption and activation of CO₂; their crystal structure provides large proportion of low-coordinated metal atoms and thus, more active sites that promote the formation of low oxidation state products like formic acid [25]. These materials are known for stabilizing reaction intermediates that are critical for CO₂ reduction, which in turn favors the formation of two-electron process to produce HCOOH over other products [26].
- The addition of CuO to the perovskite was critical in enhancing the selectivity towards formic acid. Copper-based materials are widely studied for their ability to facilitate the conversion of CO₂ to formate (HCOO⁻), which can then be protonated to form HCOOH [27]. The role of CuO in this system could be that of a co-catalyst, assisting in the adsorption of CO₂ and providing favorable conditions for the formation of formate intermediates compared to other products [28].
- The combination of K₃Bi₂I₉ and CuO likely results in a synergistic effect that enhances the overall photocatalytic performance and product selectivity. While the perovskite material facilitates CO₂ activation, CuO enhances the reduction process by stabilizing the intermediates and lowering the overpotentials associated with the reduction of CO₂ to formic acid [29]. This interaction not only improves the efficiency of the reduction but also ensures the stability of the catalyst during prolonged reaction times.

In summary, the enhanced selectivity towards HCOOH production in the photocatalytic CO₂ reduction using K₃Bi₂I₉

Table 1. Comparative analysis of HCOOH production using different photocatalytic systems (thin films and powders).

Photocatalytic system	Light source	Solvent	Efficiency ($\mu\text{mol h}^{-1}$)	Stability (h)	Ref.
NaTaO ₃ /Fe ₂ O ₃	UV light (254 nm)	Seawater	634	1	[30]
K ₃ Bi ₂ I ₉	Visible light ($\lambda > 400$ nm), 20 W	H ₂ O	283	1	[9]
KMgI ₃	Visible light ($\lambda > 400$ nm), 20 W	H ₂ O	68	NS	[31]
Ni-Cu _x O	Visible light ($\lambda > 400$ nm), 20 W	H ₂ O	15	NS	[32]
BiOI	Visible light ($\lambda > 400$ nm), 20 W	H ₂ O	14	5	[33]
Cu ₂ O/CuO	Visible light ($\lambda > 400$ nm), 20 W	H ₂ O	17	3	[34]
NH ₂ -C@Cu ₂ O	Visible light ($\lambda > 420$ nm), 300 W	Na ₂ SO ₃	7	5 cycles	[35]
Cs ₂ AgInCl ₆	UV light (380 nm), 5W	CH ₃ CN, H ₂ O, and H ₂ O/CH ₃ CN	3	2	[36]
K ₃ Bi ₂ I ₉ /CuO	Visible light ($\lambda > 400$ nm), 20 W	H ₂ O	190	5	<i>This work</i>

NS refers to not specified.

perovskite materials, particularly with the addition of CuO, can be explained by the unique properties of both materials.

Table 1 shows a comparative analysis of production of HCOOH from the photocatalytic CO₂ reduction. Compared to other reported photocatalytic systems, the K₃Bi₂I₉/CuO film heterostructured exhibits notable advantages. First, it operates under visible-light irradiation ($\lambda > 400$ nm), in contrast to benchmark perovskite-based materials such as NaTaO₃/Fe₂O₃ or Cs₂AgInCl₆, which require UV light for its activation. Second, the hydrogen evolution rate (190 $\mu\text{mol h}^{-1}$) is significantly higher than that of other visible-light-driven systems, including BiOI (14 $\mu\text{mol h}^{-1}$), Cu₂O/CuO (17 $\mu\text{mol h}^{-1}$), and Ni-Cu_xO (15 $\mu\text{mol h}^{-1}$); although it was surpassed by pristine K₃Bi₂I₉ (283 $\mu\text{mol h}^{-1}$) its stability was significantly enhanced. Importantly, the heterostructure also demonstrates enhanced stability, maintaining activity for up to 5 h, which is considerably longer than the 1 h typically reported for single-component halides. These results suggest that coupling K₃Bi₂I₉ with CuO effectively improves the balance between photocatalytic activity and operational stability, making it a promising candidate among bismuth-based lead-free halides.

Conclusions

The stability of a lead-free metal halide perovskite for CO₂ photoreduction was controlled by a bilayer thin film of K₃Bi₂I₉/MO (M=Ti, Cu, Zn). The perovskites were grown on mica substrates by the spin-coating technique, and were covered by TiO₂, ZnO, and CuO oxides deposited by physical (sputtering) and chemical (atomic layer deposition) techniques. The incorporation of the simple oxides improves the light absorption and stability of the K₃Bi₂I₉ perovskite. Even though the film of K₃Bi₂I₉/MO (M=Cu, Zn) showed a decrease on performance due to the mismatch of the band in

the bilayer film; it is necessary to note the benefit of giving stability to the halide perovskite, making it reusable and easy to recover. These strategies of encapsulation favored stability and efficiency of the bilayer, making a good alternative for a future optimization.

Acknowledgements

The authors thank SECIHTI for financial support of this research through the projects: Cátedras 1060 and Paradigmas y Fronteras de la Ciencia 320379. Also, the authors want to thank María Isabel Mendivil Palma for her support in the deposition of TiO₂ and ZnO films by the chemical methods.

References

- [1]. S. Yoshino, T. Takayama, Y. Yamaguchi, I. Akihide, A. Kudo, *Acc. Chem. Res.* **55**, 966 (2022).
- [2]. A. Mustafa, B.G. Lougou, Y. Shuai, Z. Wang, H. Tan, *J. Energy Chem.* **49**, 96 (2020).
- [3]. G. Sahara, O. Ishitani, *Inorg. Chem.* **54**, 5096 (2015).
- [4]. S. Zeng, P. Kar, U.K. Thakur, K. Shankar, *Nanotechnology* **29**, 052001 (2018).
- [5]. M.A. Raza, F. Li, M. Que, L. Zhu, X. Chen, *Mater. Adv.* **2**, 7187 (2021).
- [6]. X. Fu, T. Ren, S. Jiao, Z. Tian, J. Yang, Q. Li, *J. Energy Chem.* **83**, 397 (2023).
- [7]. Z. Chen, Y. Hu, Y. Wang, Q. Shen, Y. Zhang, C. Ding, Y. Bai, G. Jiang, Z. Li, N. Gaponik, *Chem. Mater.* **32**, 1517 (2020).
- [8]. E. Luévano-Hipólito, O.L. Quintero-Lizárraga, L.M. Torres-Martínez, *Catalysts* **12**, 1410 (2022).
- [9]. O.L. Quintero-Lizárraga, E. Luévano-Hipólito, L.I. Ibarra-Rodríguez, L.M. Torres-Martínez, *Sustainability* **15**, 16835 (2023).
- [10]. S. Chu, Y. Wang, Y. Guo, J. Feng, C. Wang, W. Luo, X. Fan, Z. Zou, *ACS Catal.* **3**, 912 (2013).
- [11]. Z.L. Liu, R.R. Liu, Y.F. Mu, Y.X. Feng, G.X. Dong, M. Zhang, T.B. Lu, *Sol. RRL* **5**, 2000691 (2021).
- [12]. Y.X. Feng, G.X. Dong, K. Su, Z.L. Liu, W. Zhang, M. Zhang, T.B. Lu, *J. Energy Chem.* **69**, 348 (2022).

- [13]. Z. Zhang, M. Wang, Z. Chi, W. Li, H. Yu, N. Yang, H. Yu, *Appl. Catal. B* **313**, 121426 (2022).
- [14]. Q.M. Sun, J.J. Xu, F.F. Tao, W. Ye, C. Zhou, J.H. He, J.M. Lu, *Angew. Chem.* **134**, e202200872 (2022).
- [15]. D. Li, Z. Xing, X. Meng, X. Hu, T. Hu, Y. Chen, *Nano Lett.* **23**, 3484 (2023).
- [16]. R. Dash, C. Mahender, P.K. Sahoo, A. Soam, *Mater. Today* **41**, 161 (2021).
- [17]. A.E. Nogueira, G.T.S.T. da Silva, J.A. Oliveira, J.A. Torres, M.G.S. da Silva, M. Carmo, C. Ribeiro, *Catal. Commun.* **137**, 105929 (2020).
- [18]. E. Luévano-Hipólito, L.M. Torres-Martínez, *Mat. Sci. Eng. B* **226**, 223 (2017).
- [19]. M.R. Alfaro Cruz, O. Ceballos-Sanchez, E. Luévano-Hipólito, L.M. Torres-Martínez, *Int. J. Hydrogen Energy* **43**, 10301 (2018).
- [20]. M.R. Alfaro Cruz, D. Sanchez-Martínez, L.M. Torres-Martínez, *Mat. Res. Bull.* **122**, 110678 (2020).
- [21]. M.R. Alfaro Cruz, D. Sanchez-Martínez, L.M. Torres-Martínez, *Int. J. Hydrogen Energy* **44**, 20017 (2019).
- [22]. J.R. Castillo-Saenz, N. Nedev, B. Valdez-Salas, M.A. Martínez-Puente, F.S. Aguirre-Tostado, M.I. Mendivil-Palma, D. Mateos, M.A. Curiel-Alvarez, O. Perez-Landeros, E. Martinez-Guerra, *J. Mater. Sci.: Mater. Electron* **32**, 20274 (2021).
- [23]. M.A. Ávila-López, E. Luévano-Hipólito, L.M. Torres-Martínez, *J. Photochem. Photobiol. A* **382**, 111933 (2019).
- [24]. W.N.R.W. Isahak, Z.A.C. Ramli, M.W. Ismail, K. Ismail, R.M. Yusop, M.W.M. Hisham, M.A. Yarmo, *J. CO₂ Util.* **2**, 8 (2013).
- [25]. H. Liu, S. Bansal, *Mater. Today Energy* **32**, 101230 (2023).
- [26]. Z. Dong, B. Li, Y. Zhu, W. Guo, *EES Catalysis* **2**, 448 (2024).
- [27]. D. Liu, Y. Hu, E. Shoko, H. Yu, T.T. Isimjan, X. Yang, *Electrochim. Acta* **365**, 137343 (2021).
- [28]. S.A. Al-Tamreh, M.H. Ibrahim, M.H. El-Naas, J. Vaes, D. Pant, A. Benamor, A. Amhamed, "Electroreduction of Carbon Dioxide into Formate: A Comprehensive Review", *ChemElectroChem* **8**, 3207 (2021).
- [29]. L.J. Minggu, M.N.I. Salehmin, M.A. Mohamed, K. Arifin, R.M. Yunus, M.B. Kassim, *Ceram. Int* **46**, 26004 (2020).
- [30]. E. Luévano-Hipólito, A. Aguirre-Astrain, L.M. Torres-Martínez, *Mater. Sci. Semicond. Proc* **185**, 108984 (2025).
- [31]. E. Luévano-Hipólito, M.G. Fabela-Cedillo, L.M. Torres-Martínez, *Mat. Lett* **361**, 136066 (2024).
- [32]. L.I. Ibarra-Rodríguez, M.R. Alfaro Cruz, L.F. Garay-Rodríguez, B.C. Hernandez-Majalca, J.L. Domínguez-Arvizu, A. López-Ortiz, L.M. Torres-Martínez, V.H. Collins-Martínez, *J. Mater. Res. Technol* **26**, 137 (2023).
- [33]. E. Luévano-Hipólito, D.A. Torres-Alvarez, L.M. Torres-Martínez, *J. Environ. Chem. Eng.* **11**, 109557 (2023).
- [34]. E. Luévano-Hipólito, L.M. Torres-Martínez, M.A. Ávila-López, *J. Phys. Chem. Solids* **170**, 110924 (2022).
- [35]. Q. Zhu, Y. Cao, Y. Tao, T. Li, Y. Zhang, H. Shang, J. Song, G. Li, *J. CO₂ Util.* **54**, 101781 (2021).
- [36]. A. Kaliyaperumal, A.V. Gopala, G. Pandurangappa, B.R.K. Nanda, R. Chetty, A.K. Chandiran, *ACS Appl. Mater. Interfaces* **17**, 13896 (2025).